In this study, the causes and mechanisms of thermal distortion and quenching crack and the effects of kinds of cooling methods and cooling conditions of reverse transformation treatment were analyzed in order to prevent these troubles by the model of metallo-thermo-mechanics (5), (6). The cooling methods considered in these analyses were immersion cooling, spray cooling and mist-spray cooling.

ANALYSIS METHOD ANALYSIS MODEL

Since the distribution of temperature and stress/strain and their transition and behavior of phase transformation affect mutually during third cooling, analyses combined distortion analysis with heat transfer analysis and analysis of phase transformation behavior are necessary to examine thermal distortion and stress generation considering these mutual effects. Therefore, the behavior of thermal distortion and stress generation were analyzed by the model of metallo-thermo-mechanics shown in Fig.1(5), (6): COSMAP (7) in this study. Thermo-elastic-plastic analysis, analysis of heat transfer and analysis about transformation behavior are combined in this model considering of thermal shrinkage and expansion, shrinkage and expansion accompanying transformation. The effects of cooling methods and cooling conditions on the behavior of thermal distortion and stress generation were studied by this model.



Fig.1 – Schematic view of the model of metallo-thermo-mechanics. (5)

ANALYSIS CONDITIONS

The analyses were performed using a finite element method two-dimensional model for total cross section of a bloom having a square cross section of 200 mm (thickness) x 200 mm (width). The analyzed steel type was case hardening steel SCr420(JIS), and the transformation behavior of this steel type was estimated in consideration of the CCT diagram measured in a coarse austenite structure similar to that of bloom as cast continuously (8).

In these calculations, the solidification calculation was not

formed, following c 1123K, and and cooling face layer o med into a stenite stru calculation until the vo came less tl assuming and the eff cross-secti tion behavi The kinds were imme ring, and wa at a water d cooling whi min)). The transfer and method an relationship in Fig.2 (9)-



RESULTS AND DISCUSSION

The cooling time required for the refinement of austenite grains in the surface layer of the bloom within a range of 10 mm by the reverse transformation treatment for various cooling methods and cooling conditions was clarified by the calculations. Fig. 3 is a bar graph showing the results of estimated the cooling time required for refining austenite grains. The required cooling time was determined as a time at which the volume fraction of austenite was 0.1 or less within a range of 10 mm from the surface of the bloom. As shown in this figure, the required cooling time decreased with the increase of the cooling strength together with the uniform Tw 0 -1.667 Td[(togTh32IrPu9ume fra

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effects the bloc were studi chanics. As contour di bainite and sion coolin cooling wit shown in section of 40 times th by cooling colors in th In the case sfer coeffic spray cooli







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Fig.10 – Distribution of Sxx and volume fraction of martensite when Sxx is maximum during uniform immersion cooling. (Water jet stirring)

(b) Volume fraction of martensite

From the viewpoint of the prevention of the occurrence of guenching cracks, the behavior of stress generation in the cooling process by each cooling method was analyzed. Fig.10(a) and Fig.11(a) show the distributions of normal stress Sxx in the cross section of bloom in the case of uniform immersion cooling with water jet stirring and air stirring, when the stress Sxx during the cooling is maximum. Fig.10(b) and Fig.11(b) show the distributions of martensite volume fraction in the cross section at that time. The Sxx became maximum in the position indicated by the circle. In these cases of uniform immersion cooling, the amount generated bainite and pearlite phase is very small, and it is considered that the transformation expansion due to martensitic transformation near the surface of bloom and thermal stress cause the generation of maximum Sxx in these cases. In the case of immersion cooling with water jet stirring (Fig.10), the Sxx became maximum in the austenitic phase region just below the surface at the center of the surface. It was seemed that the Fig.11 – Distribution of Sxx and volume fraction of martensite when Sxx is maximum during uniform immersion cooling. (Air stirring)

(b) volume fraction or martensite

stress Sxx became the maximum just below the center of the surface, because the austenitic phase region just below the surface at the width center was pulled in the bloom width direction by the expansion due to martensitic transformation at the surrounding surface. On the other hand, in the case of immersion cooling with air stirring (Fig.11), the surface layer at the off-corner was pulled in the bloom width direction due to the progress of martensitic transformation inside the cross section of the corner. It is considered that the maximum stress Sxx occurred at the position, because of the martensitic transformation, the effect of thermal shrinkage and the large deformation resistance at the region caused by the low temperature.



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layers and increased Sxx at the surface.

When the water density on the bloom upper surface side is 20 (l/(m²• min)), and the other three surfaces are cooled by spray at 100 (l/(m²• min)), distribution of Sxx and martensite volume fraction in the cross section of bloom are shown in Fig.14 in comparison with the case of cooling uniformly on all four sides with water density of 100 (l/(m²• min))(Fig.13). In the case of uniform spray cooling with 100 (l/(m²• min) shown in Fig.13, the maximum Sxx occurred on the off-corner surface due to expansion accompanying martensitic transformation at inside near the corner. In the non-uniform spray cooling (Fig.14), the upper surface layer whose temperature is higher than the other three surface layers is pulled due to martensitic transformation inside the cross section and the thermal shrinkage caused by constraint from the side surface of which temperature is lower. For the above reasons, it was found that the largest Sxx occurred at the center of the width of the upper surface in this case.

Compared with uniform cooling, in non-uniform cooling, the position, timing, and mechanism at which the maximum Sxx occurs differ greatly, and the maximum Sxx greatly increased in the case of non-uniform cooling with strong cooling intensity.

The maximum Sxx during cooling in various cooling methods and conditions is shown in the bar graph of Fig.15. In the uniform immersion cooling, the maximum Sxx was almost constant regardless of the stirring power of the immersion bath. In the uniform cooling with spray and mist-spray, Sxx decreased as the water density decreased. In this study, in spray cooling and mist cooling, there are some cases where non-uniform cooling significantly increased the maximum Sxx in comparison with uniform cooling.

From the results shown in this figure, it was found that spray cooling or mist cooling at a low water density of about 20 (l/(m²• min)) was preferable to reduce the stress of Sxx etc. It was estimated that it was important for decrease of stress and prevention of quenching crack occurrence to reduce the variation in cooling strength between surfaces of bloom.



Fig.15 – Max Sxx in the case of each cooling method and each cooling condition.

SAMMARY AND CONCLUSION

In this study, the causes of thermal distortion and quenching crack and the effects of kinds of cooling methods and cooling conditions in the reverse transformation treatment were analyzed in order to prevent these troubles by the model of metallo-thermo-mechanics. The cooling methods considered in these analyses were immersion cooling, spray cooling and mist-spray cooling. The following results were obtained from these analyses.

1) In several kinds of cooling methods (immersion cooling, spray cooling and mist-spray cooling) and various cooling conditions, the cooling time required for the reverse transformation treatment were clarified.

2) The effects of the cooling methods and various cooling conditions on the cross-sectional shape of the cast bloom after cooling and the stress generated during co-